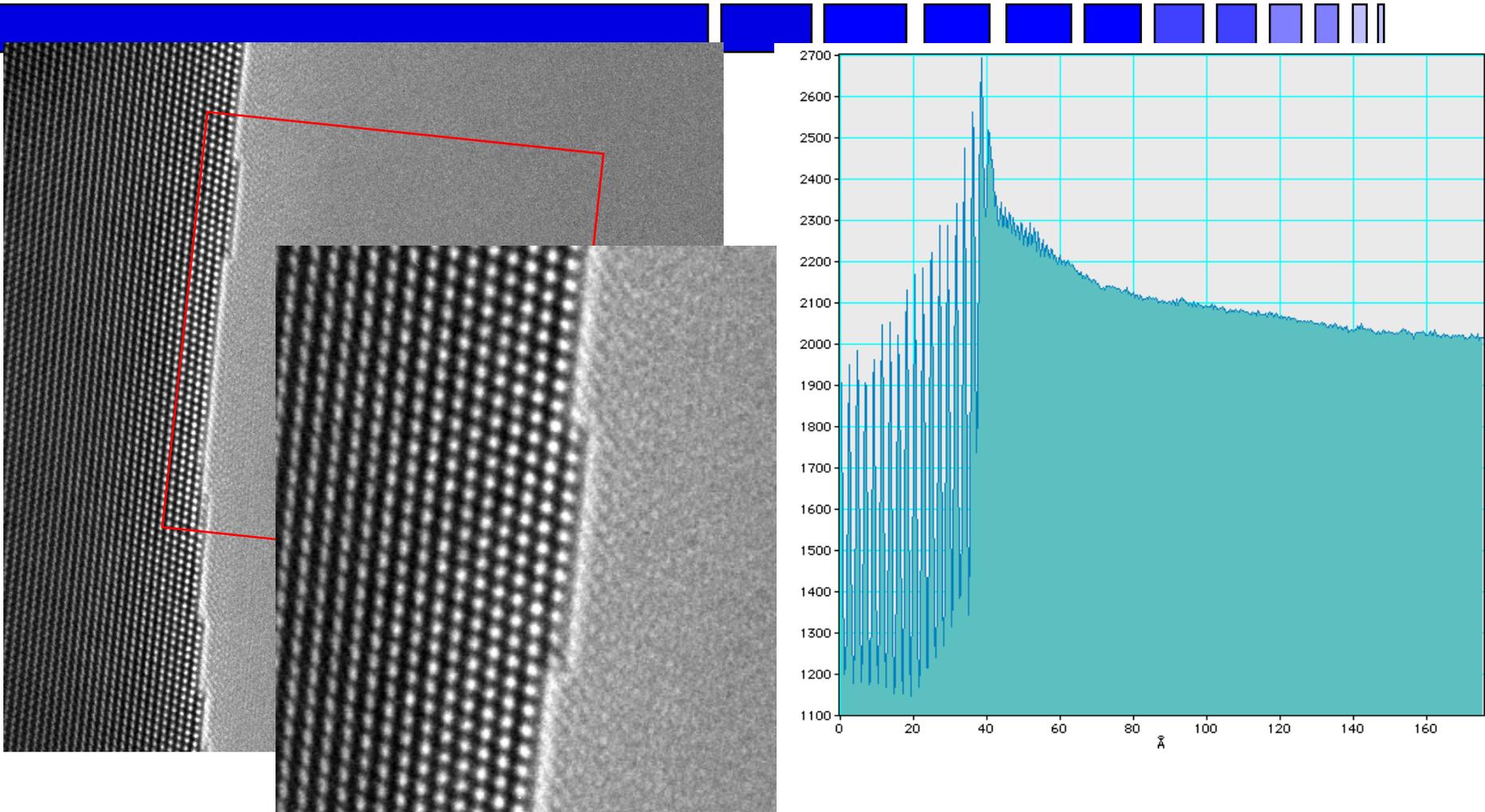


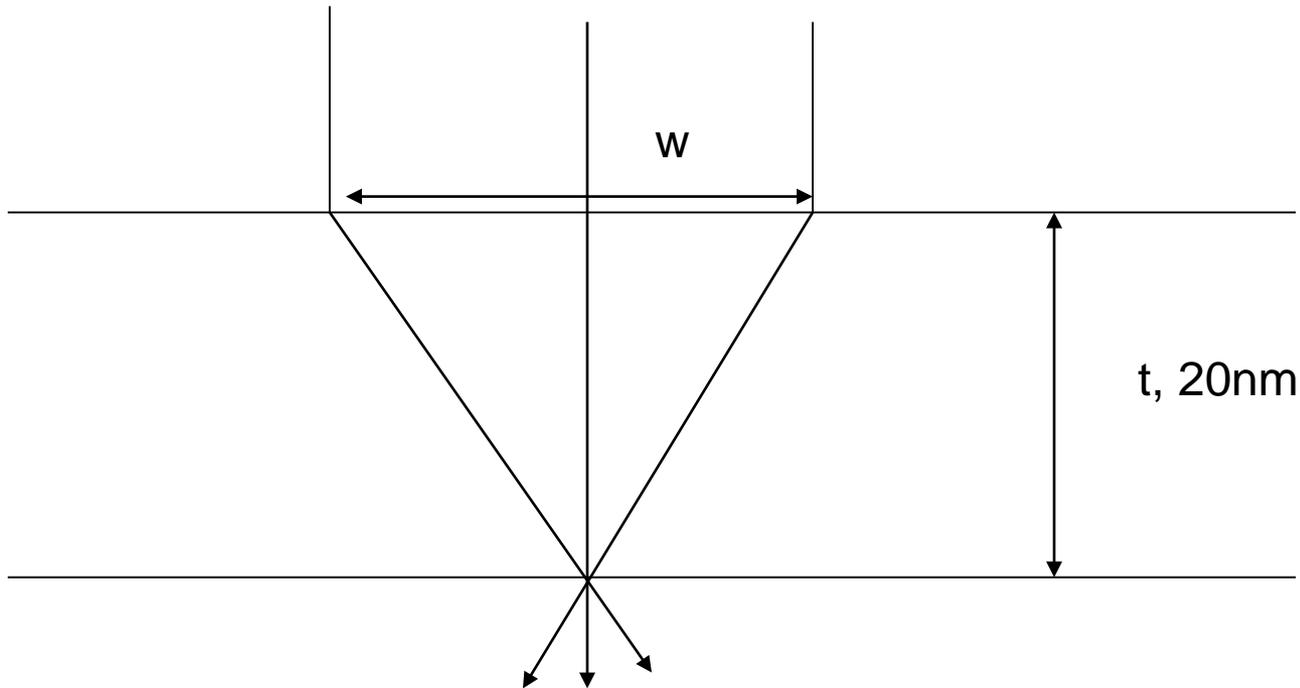
Atomic resolution micrograph of multiply-twinned nanocrystalline film of Si. (C. Song)

# Au [110] – Vacuum wave



Courtesy C. Kisielowski, J.R. Jinschek (NCEM, Berkeley)

# But...column approximation?



$w = 2t\theta$ . Take  $\theta=10$  mRad,  $t=20$ nm,  $w=4$  Angstroms

# Basis Sets

- 
- Plane waves
    - Solutions in vacuum
    - Used in kinematical theory
  - Bloch waves
    - Used in 2-beam theory (and beyond)
    - Combinations of plane waves
  - Tight-binding model
    - Uses atom-like orbitals (similar to chemistry, MO)

# Basis Sets Continued

- 
- What to use?
  - No simple answer
  - The one which gives the simplest explanation with the least complications (KISS = Occam's Razor) is normally best
  - Often this changes depending upon the type of problem/property of interest...

# Channeling: Real-Space model

- Electron channeling approximation\*:
  - 1) Ignore all but ZOLZ interactions (high energy)
  - 2) Assume sample orientation so that one has well separated atomic columns along  $z$  (in initial expansion only)
  - 3) Use atomic-column based tight-binding model

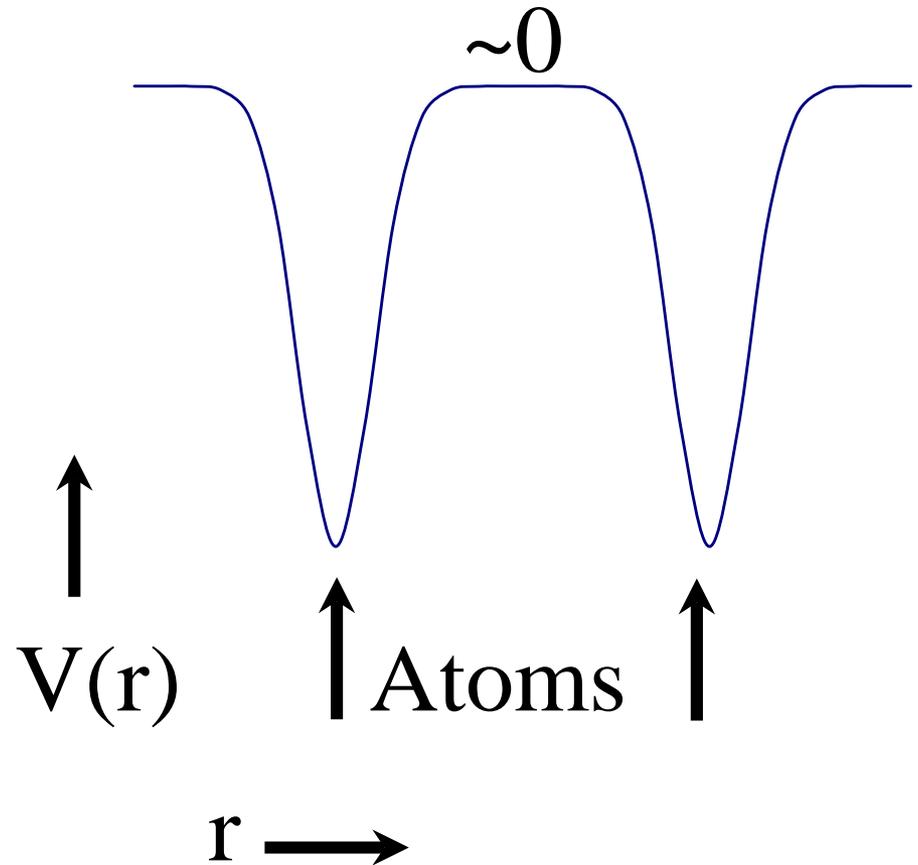
\* Fujimoto, phys. stat. sol. (1978) + many others

Van Dyck and Op de Beeck, *Ultramicroscopy* (1996)

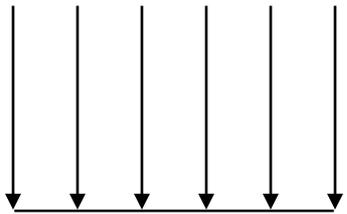
# Channeling Concept



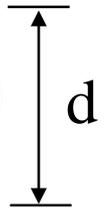
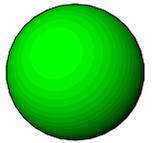
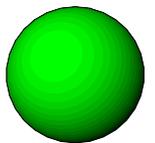
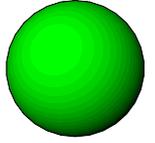
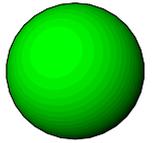
- For many zone axes, in projection atoms are well separated
- Potential large near columns
- Small between columns



# Consider an isolated column: I



$e^-$



$$V(\underline{R}) = V(x, y) = \frac{1}{d} \int_{-d/2}^{d/2} V_0(\underline{R}, z) dz$$

Average potential

$$\nabla^2 \psi(\mathbf{r}) + \frac{2m}{\hbar^2} [E + V(\mathbf{r})] \psi(\mathbf{r}) = 0$$



$$\nabla^2 \psi(\mathbf{r}) = \nabla_R^2 \psi(\mathbf{r}) + \partial^2 \psi(\mathbf{r}) / \partial z^2$$

$$\psi(\mathbf{r}) = \exp(-2\pi i k z) \Omega(R)$$

$$\nabla^2 \psi(\mathbf{r}) = \exp(-2\pi i k z) \{ \nabla_R^2 \Omega(\mathbf{r}) + \cancel{\partial^2 \Omega(\mathbf{r}) / \partial z^2} - 4\pi i k \partial \Omega(\mathbf{r}) / \partial z - 4\pi^2 k^2 \Omega(\mathbf{r}) \}$$

$$\nabla^2 \psi(\mathbf{r}) = \exp(-2\pi i k z) \{ \nabla_R^2 \Omega(\mathbf{r}) - \overset{\updownarrow}{E} \Omega(\mathbf{r}) - 4\pi i k \partial \Omega(\mathbf{r}) / \partial z \}$$

$$\nabla_R^2 \Omega(\mathbf{r}) - 4\pi i k \partial \Omega(\mathbf{r}) / \partial z + \frac{2m}{\hbar^2} [V(\mathbf{r})] \Omega(\mathbf{r}) = 0$$



$$\nabla_R^2 \Omega(\mathbf{r}) - 4\pi i k \partial \Omega(\mathbf{r}) / \partial z + \frac{2m}{\hbar^2} [V(\mathbf{r})] \Omega(\mathbf{r}) = 0$$

$$4\pi i \partial \Omega(\mathbf{r}) / \partial z = (1/k) \nabla_R^2 \Omega(\mathbf{r}) \quad \text{Spreading of information}$$

$$4\pi i \partial \Omega(\mathbf{r}) / \partial z = -\frac{2m}{\hbar^2 k} [V(\mathbf{r})] \Omega(\mathbf{r}) \quad \text{Diffraction}$$



$$\nabla_R^2 \Omega(\mathbf{r}) - 4\pi i k \partial \Omega(\mathbf{r}) / \partial z + \frac{2m}{\hbar^2} [V(\mathbf{r})] \Omega(\mathbf{r}) = 0$$

$$\Omega(\mathbf{r}) = (m / 2\pi k \hbar^2) \Phi(R) \exp(-i\varepsilon z)$$

$$\nabla_R^2 \Phi(\mathbf{R}) + \frac{2m}{\hbar^2} [\varepsilon + V(\mathbf{R})] \Phi(\mathbf{R}) = 0$$

# Consider an isolated column: II



- Problem reduces to 2-D Schrödinger's Eqn

$$\nabla_{\mathbf{R}}^2 \Phi(\mathbf{R}) + \frac{2m}{\hbar^2} [\varepsilon + V(\mathbf{R})] \Phi(\mathbf{R}) = 0$$

- Solutions have form:

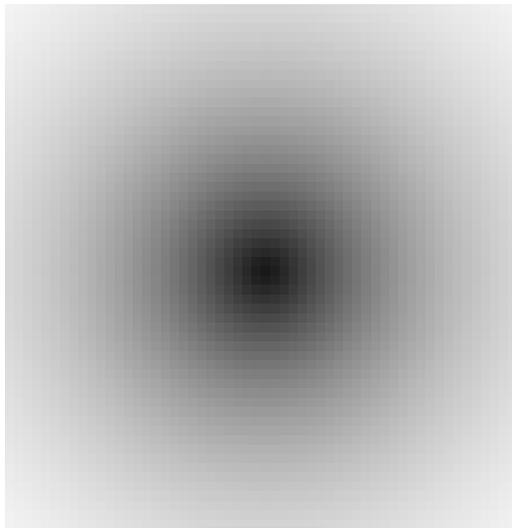
$$\psi(\mathbf{r}) = \sum_n C_n \Phi_n(\mathbf{R}) \exp\{-i\pi\varepsilon_n z\}$$

↑  
2D orbitals

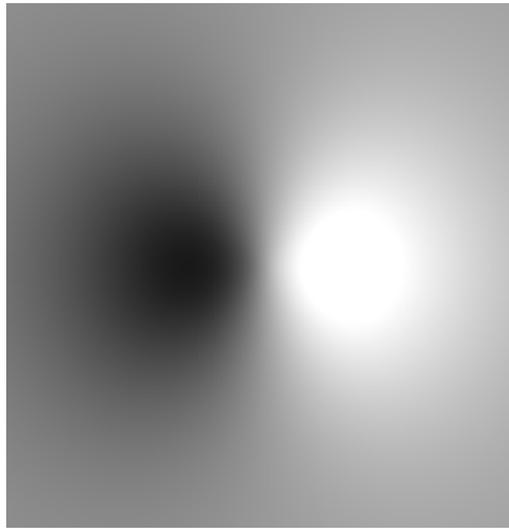
“Transverse Energy”  
↙

# Character of States

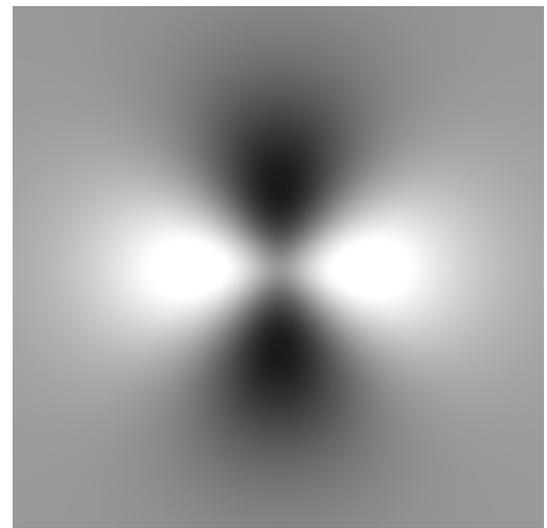
- 2D analogues of 3D atomic orbitals



1s



2p



3d

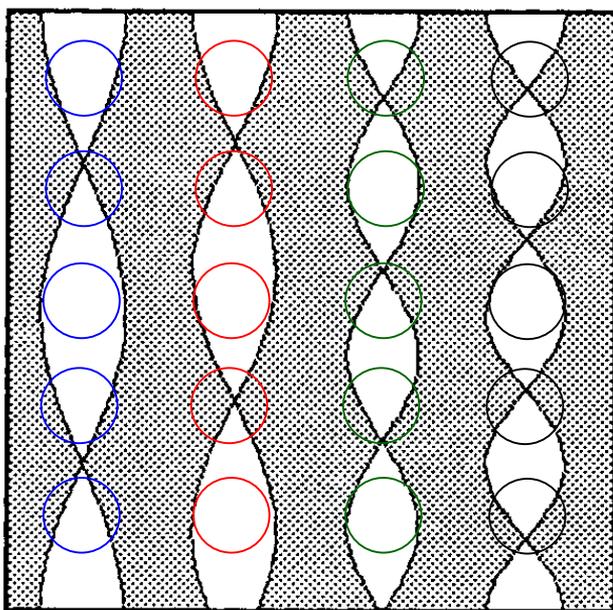
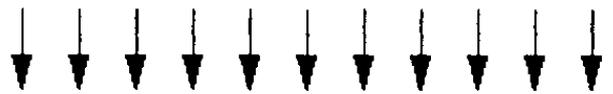
# Dynamical Solutions: many columns

- General solution

$$\psi(\mathbf{R}, z) - 1 \approx \sum_{i,j} C_i \Phi_i(\mathbf{R} - \mathbf{R}_j) (\exp(-i\pi\varepsilon_n z) - 1)$$

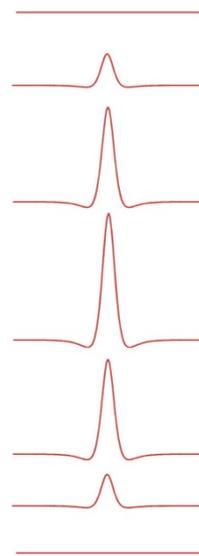
- On a zone (symmetry), s-type orbitals dominate
- Not so simple if the zone axis is complicated – not a solved problem

$$\psi(\mathbf{R}, z) - 1 \approx \sum_{i,j} C_i \Phi_i(\mathbf{R} - \mathbf{R}_j) (\exp(-i\pi\varepsilon_n z) - 1)$$

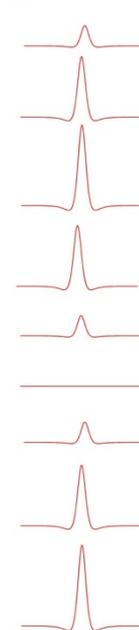
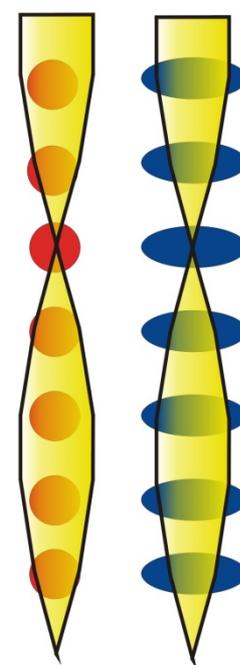


light atoms

heavy atoms



light atoms



heavy atoms

Curtesy D. Van Dyck

# A Quick Comparison



- Bloch Wave

- Different States
- Oscillate with Depth

- Interference

- Thickness Fringes, etc

- Tight Binding

- Atomic States
- Oscillate with Depth

- Interference

- Atomic contrast

Note: ignore overlap of states

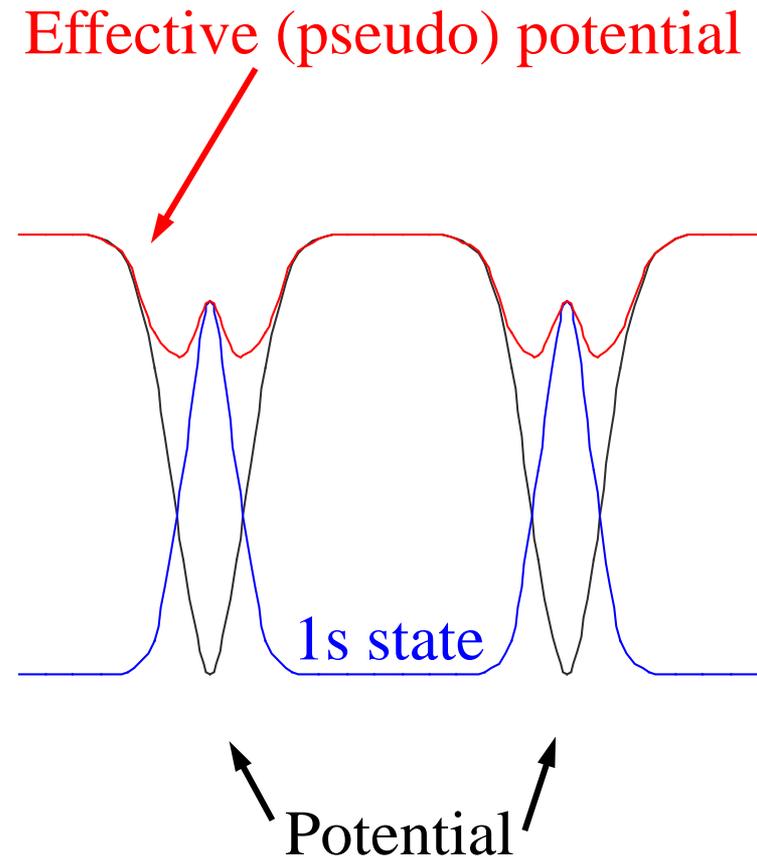
# OPW

- Suppose we have some atomic-like state,  $\Phi(r)$  and also add a plane-wave like term
- The two must be orthogonal – a plane wave is not
- Instead use
$$\exp(2\pi i k \cdot r) - \Phi(r) \int \Phi(r)^* \exp(2\pi i k \cdot r) dr$$
- This is small when  $\Phi(r)$  is large, misses the core

# So what does this gain us?

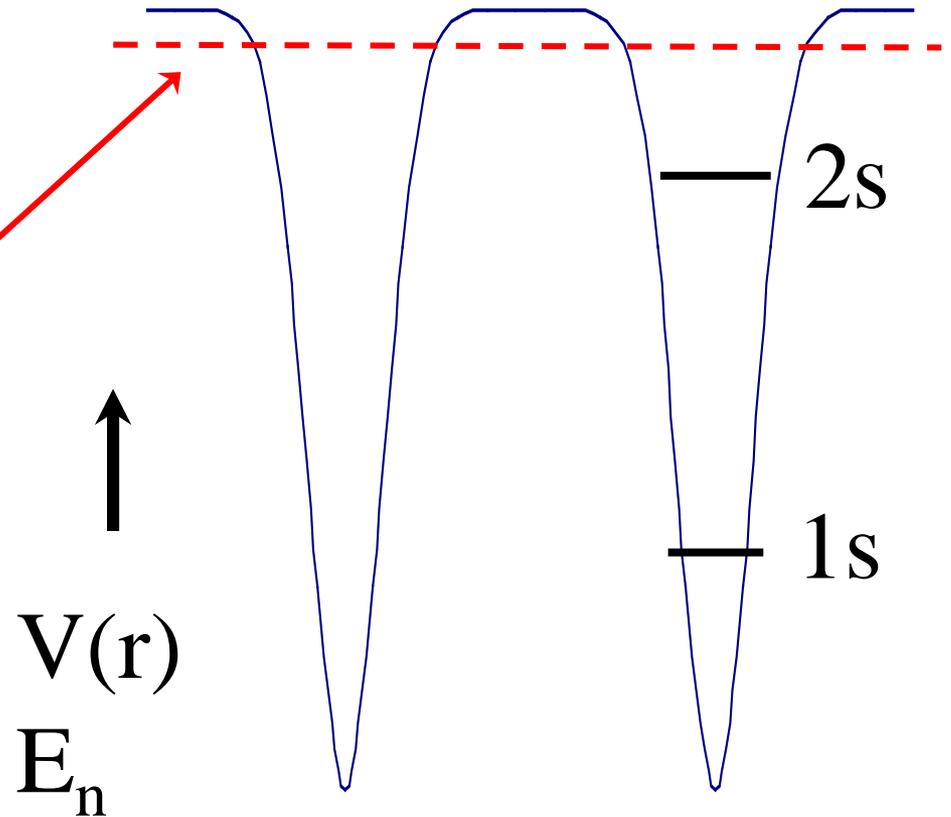
- First core state (1s) cancels most of potential (OPW)
- Only need to consider a few states, unlike classic Bloch-wave methods

N.B. Tight-binding

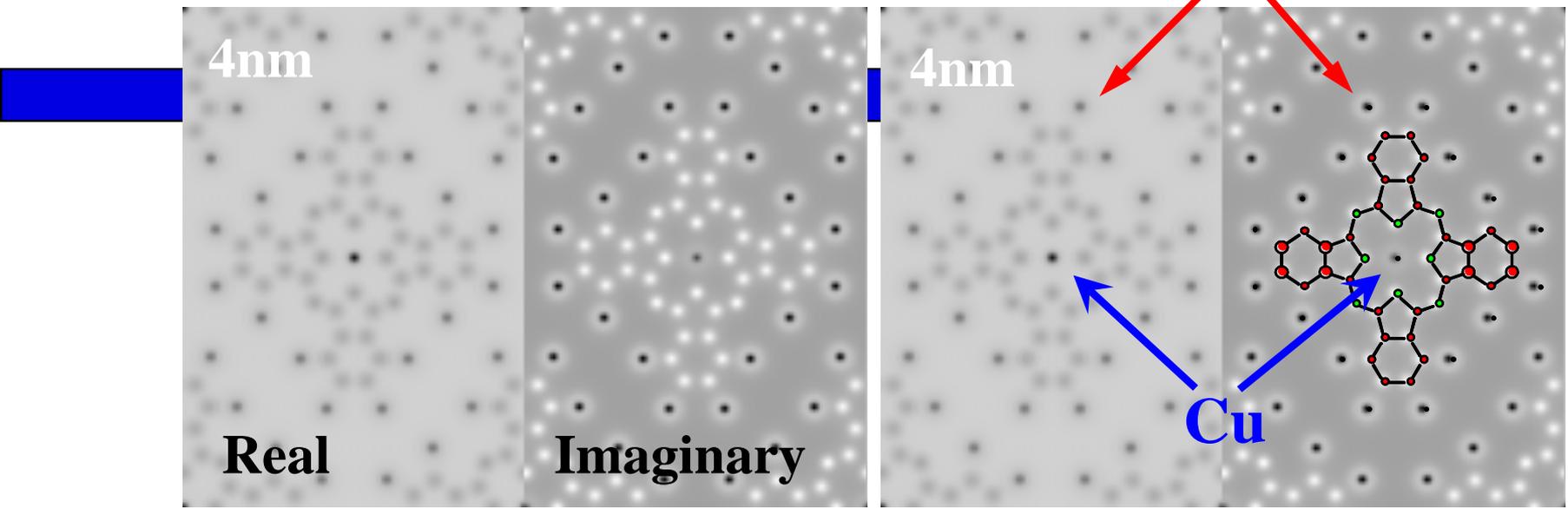


# Channeling Concept

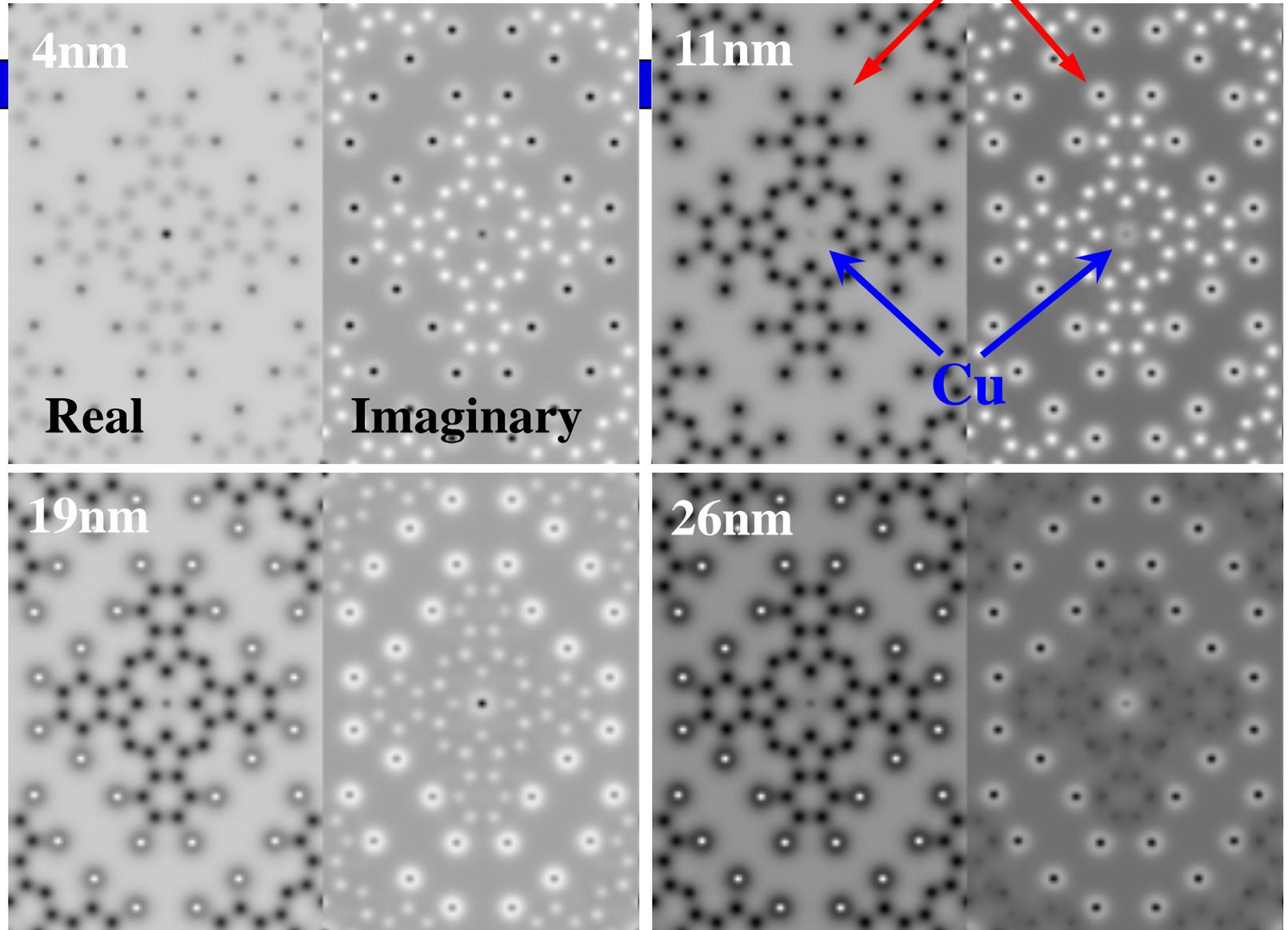
- Lowest Transverse Energy states (1s) dominate
- Can ignore (to 2<sup>nd</sup> order) delocalized states



# Full wave



# Full wave



# Dynamical Solutions: many columns

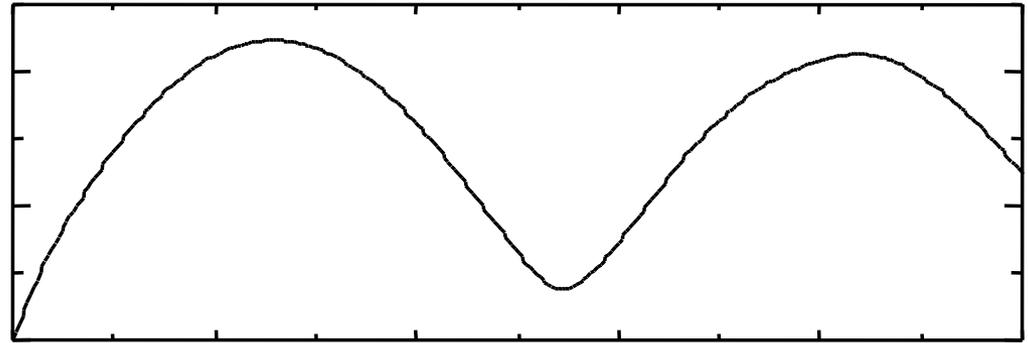

$$\psi(\mathbf{R}, z) - 1 \approx \sum_{i,j} C_i \Phi_i(\mathbf{R} - \mathbf{R}_j) (\exp(-i\pi\varepsilon_n z) - 1)$$

$$|\psi(\mathbf{R}, z) - 1|$$

Peaked atom-like term, localized at columns in projection

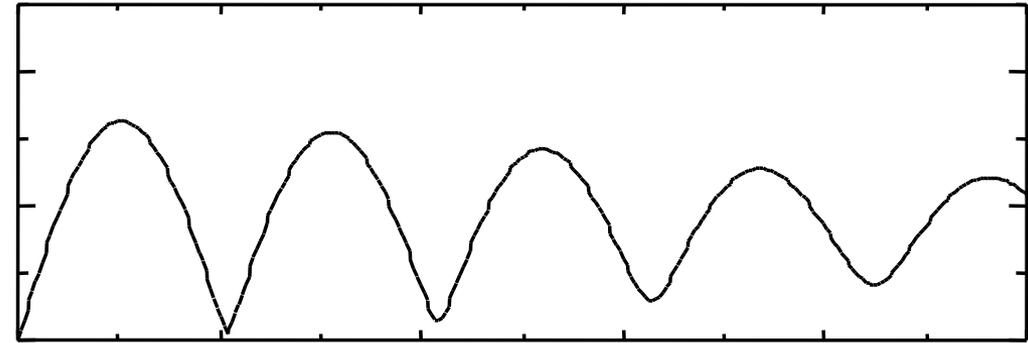
# Depth dependence by atom type

O

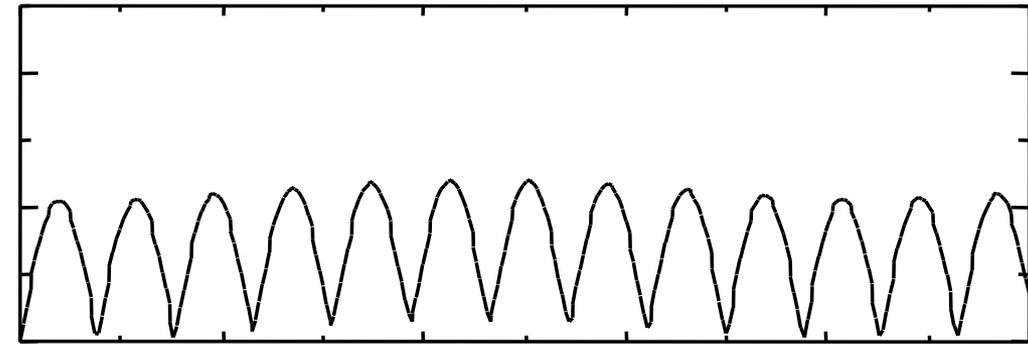


$|1-\psi(r)|$   
at atom

Al



Ga



Thickness [Å]

# Why?


$$\psi(\mathbf{R}, z) - 1 \approx \sum_{i,j} C_i \Phi_i(\mathbf{R} - \mathbf{R}_j) (\exp(-i\pi\varepsilon_n z) - 1)$$

$z$  small

$$\psi(\mathbf{R}, z) - 1 \approx \sum_{i,j} C_i \Phi_i(\mathbf{R} - \mathbf{R}_j) (-i\pi\varepsilon_n z)$$

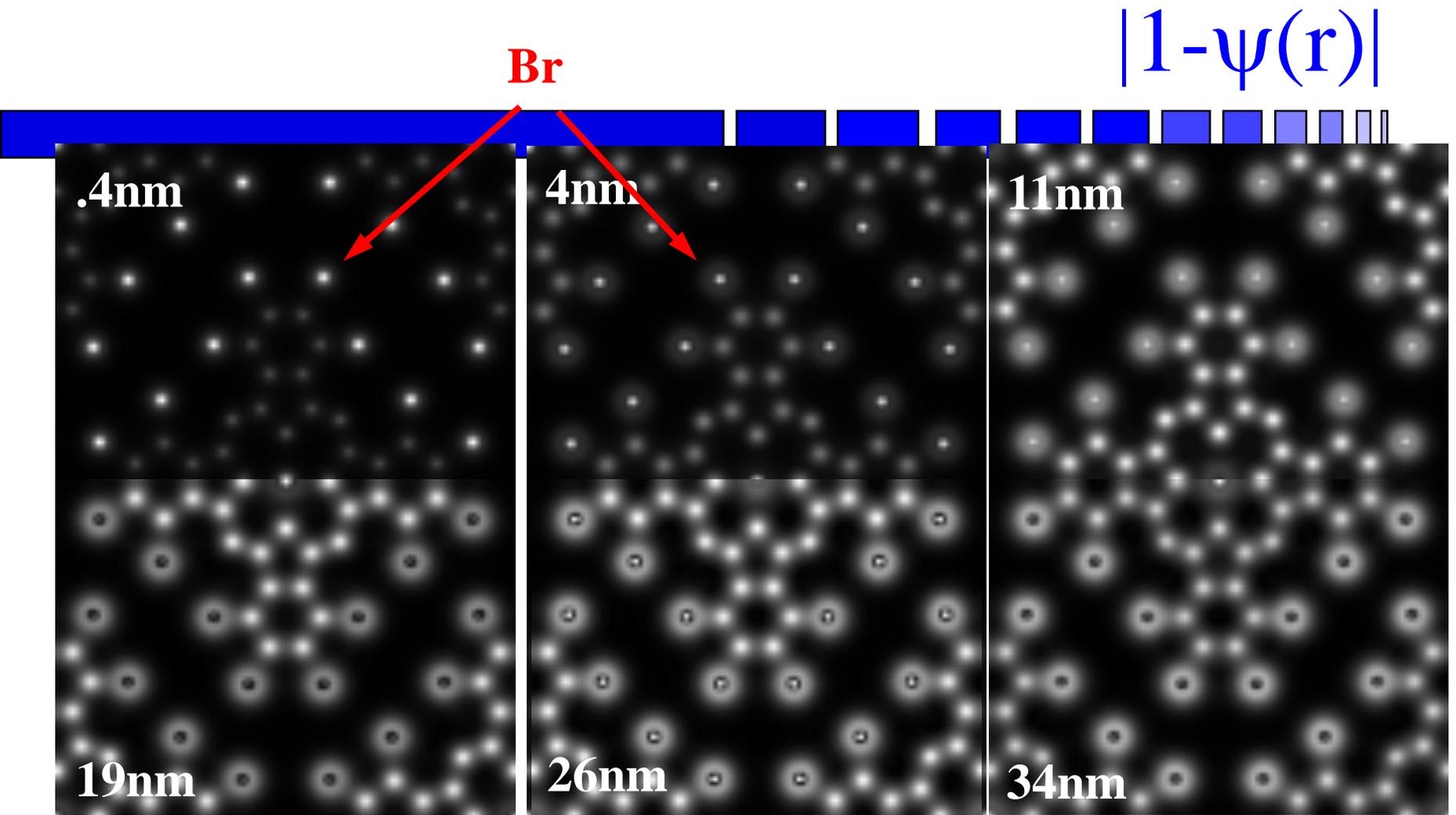
Phase grating approximation

$$\psi(\mathbf{R}, z) \approx \exp(i\sigma z V(\mathbf{R}))$$

$$\psi(\mathbf{R}, z) - 1 \approx i\sigma z V(\mathbf{R})$$

$$C \approx \sigma / \varepsilon_n$$

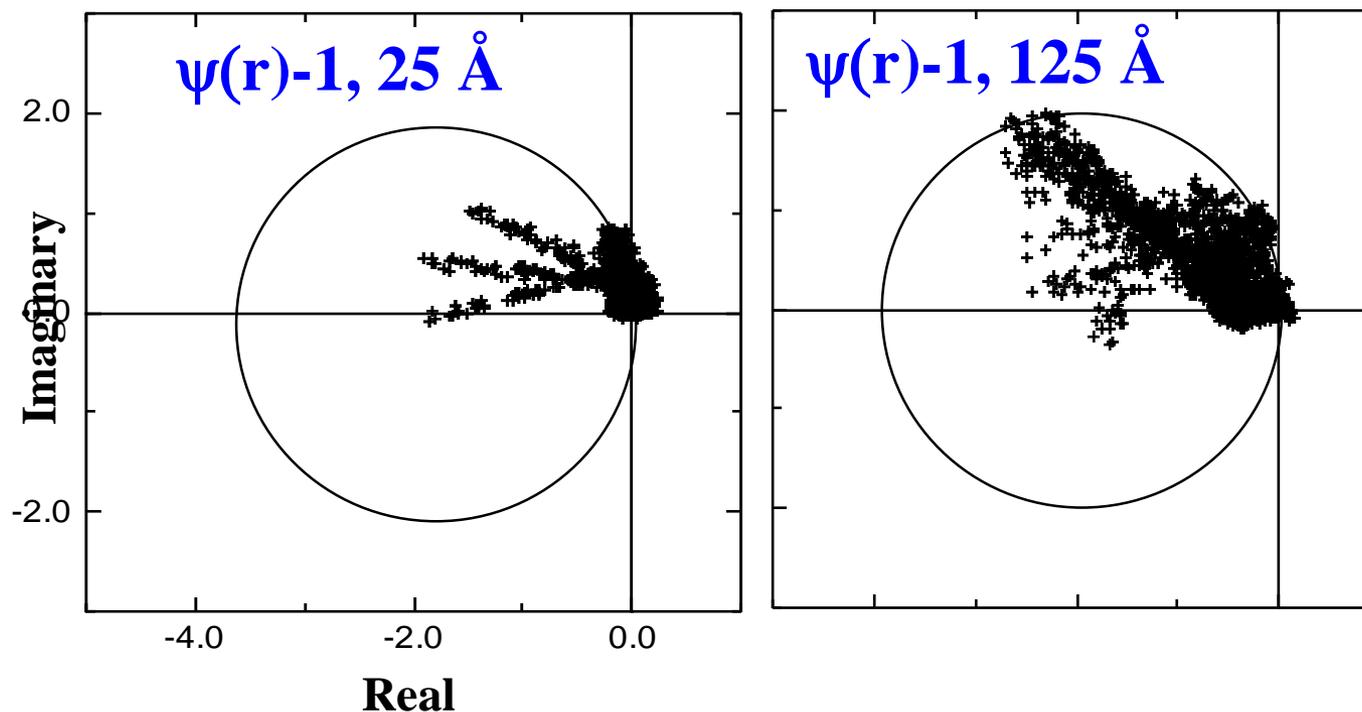
$\varepsilon_n$  is larger for heavier atom



Carbon is more important than Bromine for  $\sim 10\text{nm}$

# Sinkler Plots: Quasi-Kinematical

$\psi(r)-1$  has atom-like peaks with species-dependent oscillation

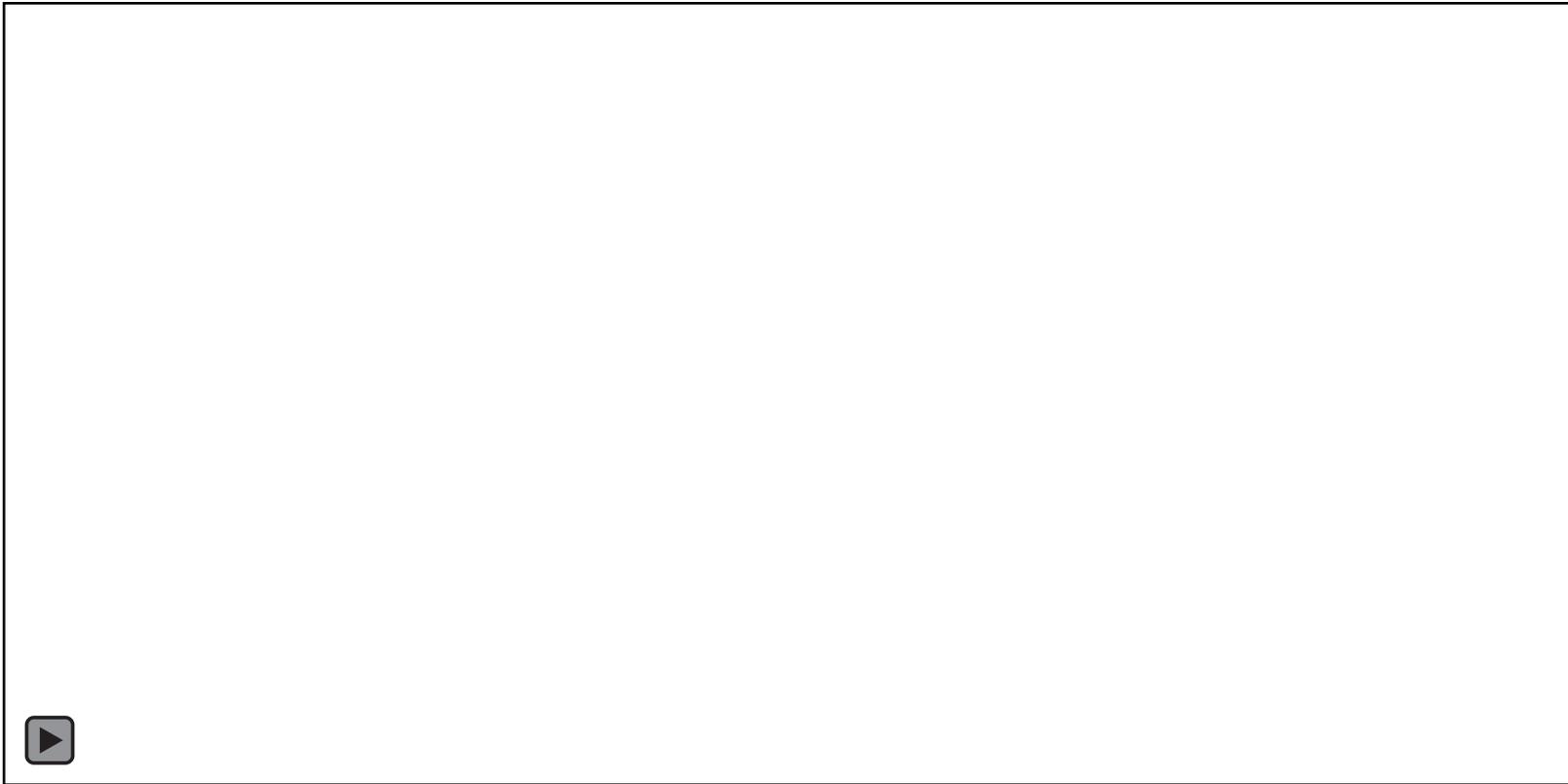


Note: if the points lay on a straight line the result is pseudo-kinematical, i.e. it walks and talks like kinematical even though it is not

# Thickness Movie



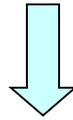
Oxygen 1s States



Metal 1s States

# Bloch-wave formulation (dynamical theory)

Schroedinger equation



Wave function of the incident electron:

$$\Psi(\mathbf{r}) = \sum_j \varepsilon^{(j)} b^{(j)}(\mathbf{r})$$

Bloch state (eigen state):

$$b^{(i)}(\mathbf{r}) = \sum_{\mathbf{g}} C_{\mathbf{g}}^{(i)} \exp[2\pi i(\mathbf{k}^{(i)} + \mathbf{g}) \cdot \mathbf{r}]$$

- Electron density distribution of the Bloch state:  $|b^{(i)}(\mathbf{r})|^2$
- Excitation of the Bloch states:  $\varepsilon^{(i)}$

# Dispersion surface and Excitations of Bloch states SrTiO<sub>3</sub> [001]

Dispersion surface

$k_z$

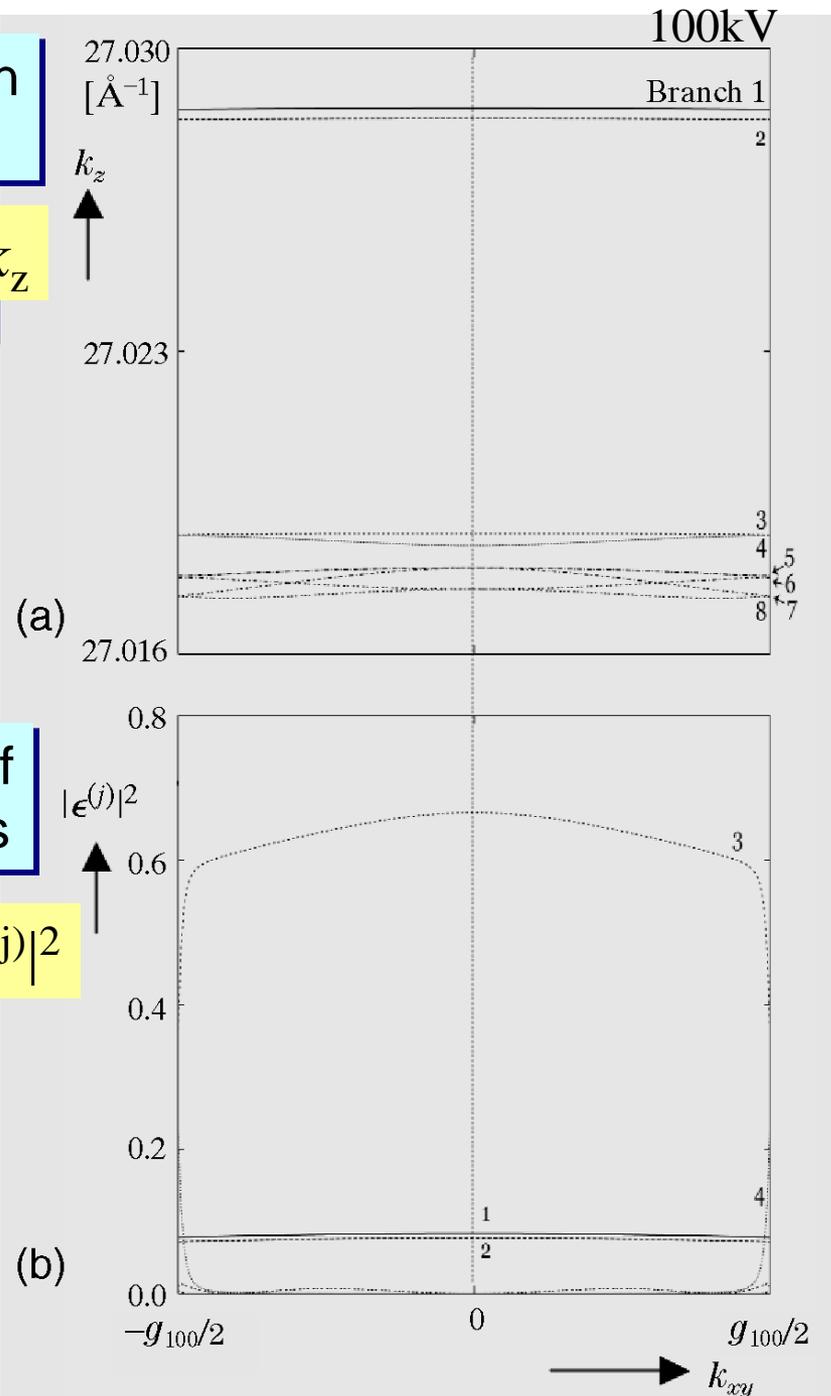
Branches with lower potential energy  
 → Higher kinetic energy ( $k$ )

Excitations of the branches

$|\epsilon^{(j)}|^2$

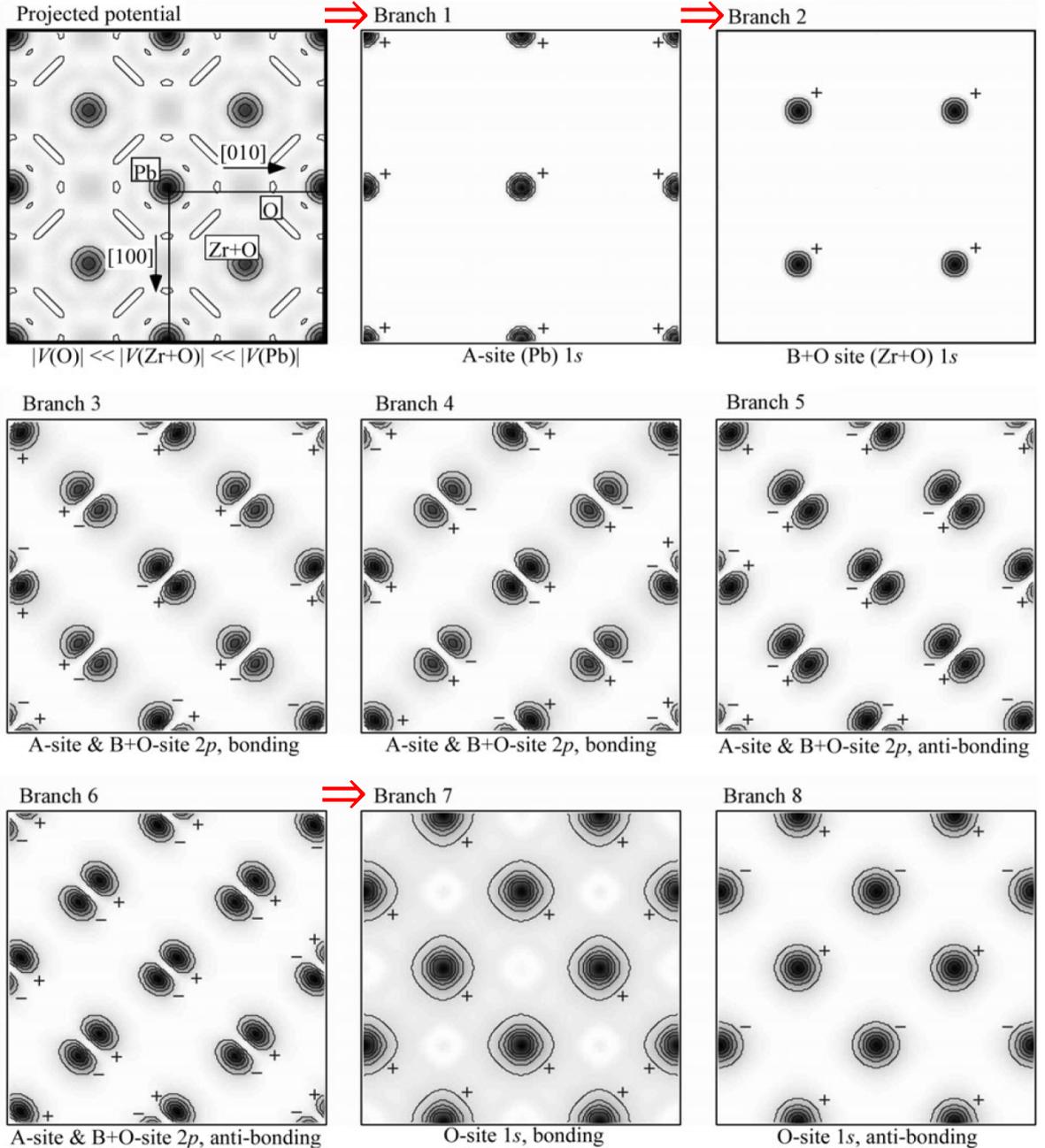
(Energy conservation)

Courtesy of K. Tsuda



# PbZrO<sub>3</sub> [001]

## Bloch-state simulations



$\Rightarrow$  : well-excited  
at ZA

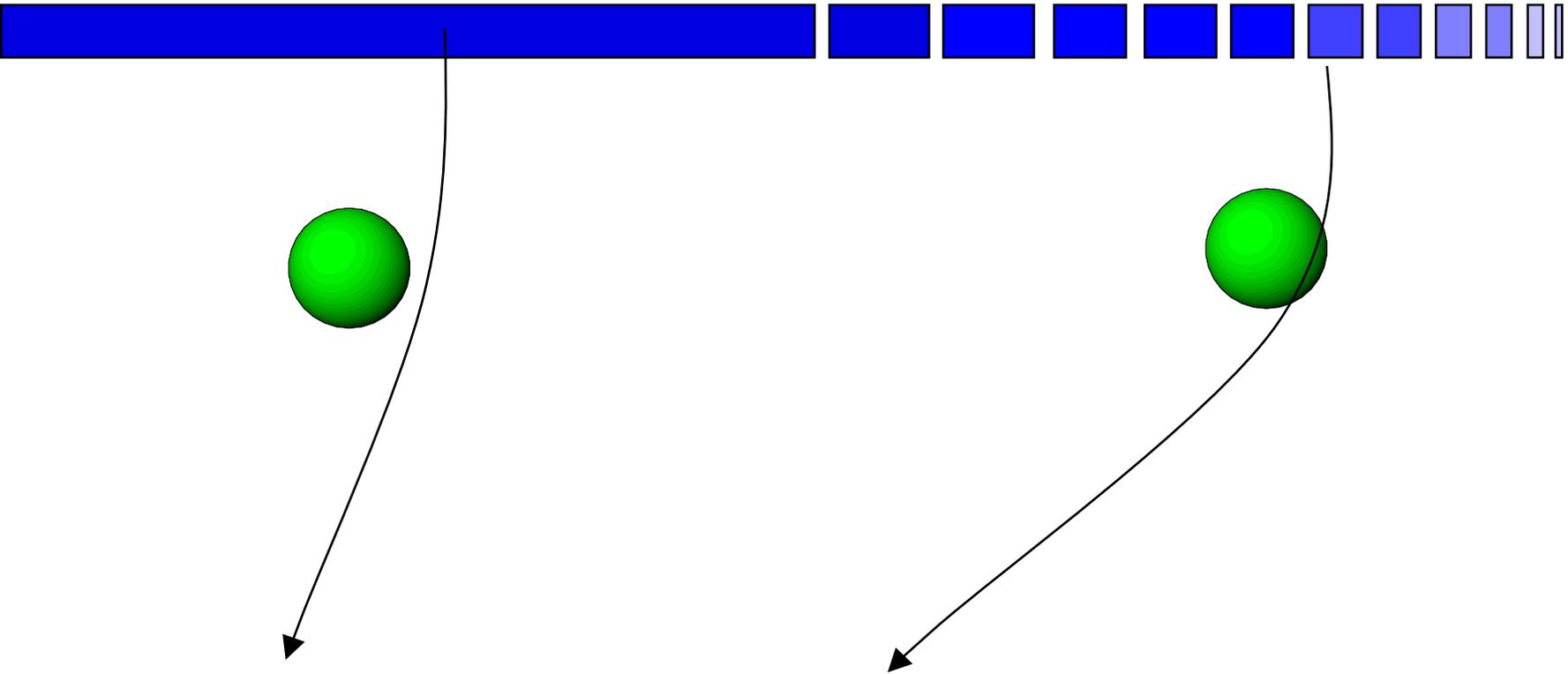
# Caveats

- 
- In a thick crystal, must include all states
  - Higher  $E_n$  states not so “atomic”
  - Overlap of states on different columns can be important
  - Tilt off zone axis  $\rightarrow$  excites more p, d states
  - Strain=tilt, hence may destroy simple picture

# Relevance 1: HREM

- 
- For a thin sample the 1s states dominate – simple
  - The scattering is pseudo-kinematical, easy to interpret
  - In thicker samples ( $> 20\text{nm}$ ) not quite so simple

# Relevance 2: ADF

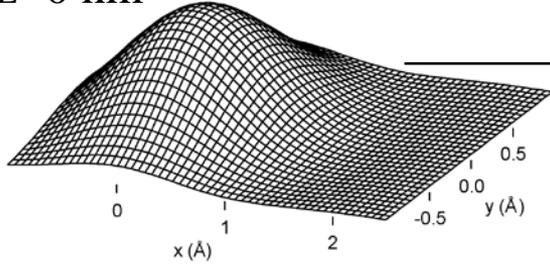


The closer the electron comes to the nucleus, the higher the probability of high-angle scattering (elastic or inelastic)

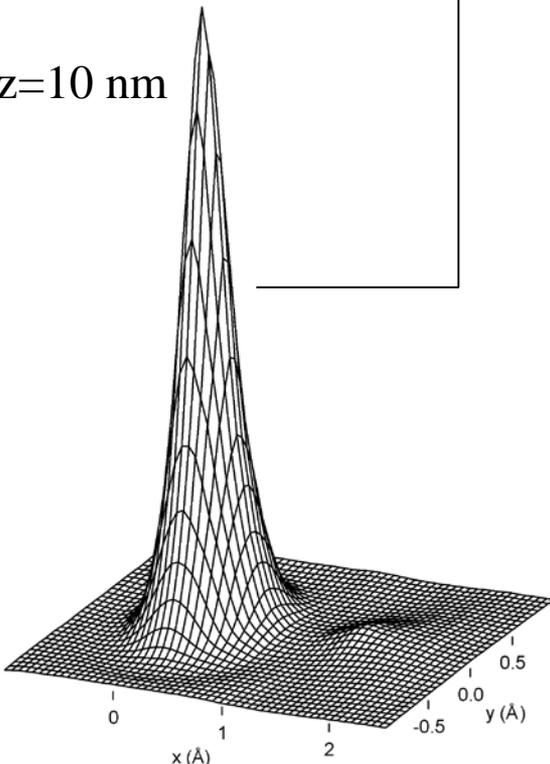
1s states dominate the high-angle scattering

# Probe Channeling

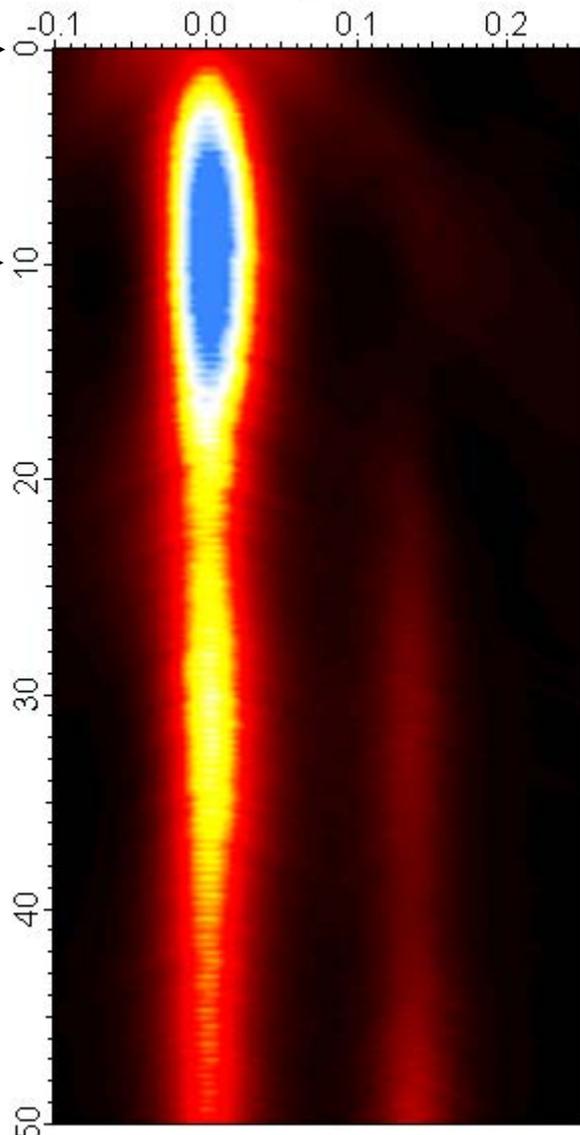
$z=0$  nm



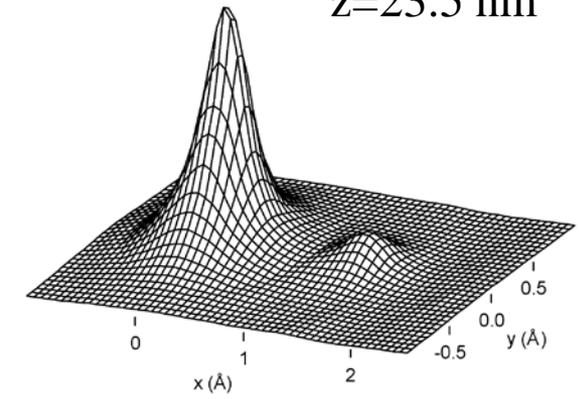
$z=10$  nm



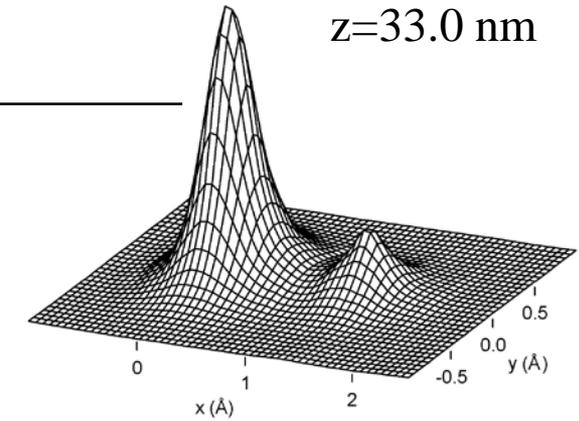
●  $x$  (nm) ●



$z=23.5$  nm



$z=33.0$  nm



# Strain Contrast at Si/SiO<sub>2</sub> Interfaces

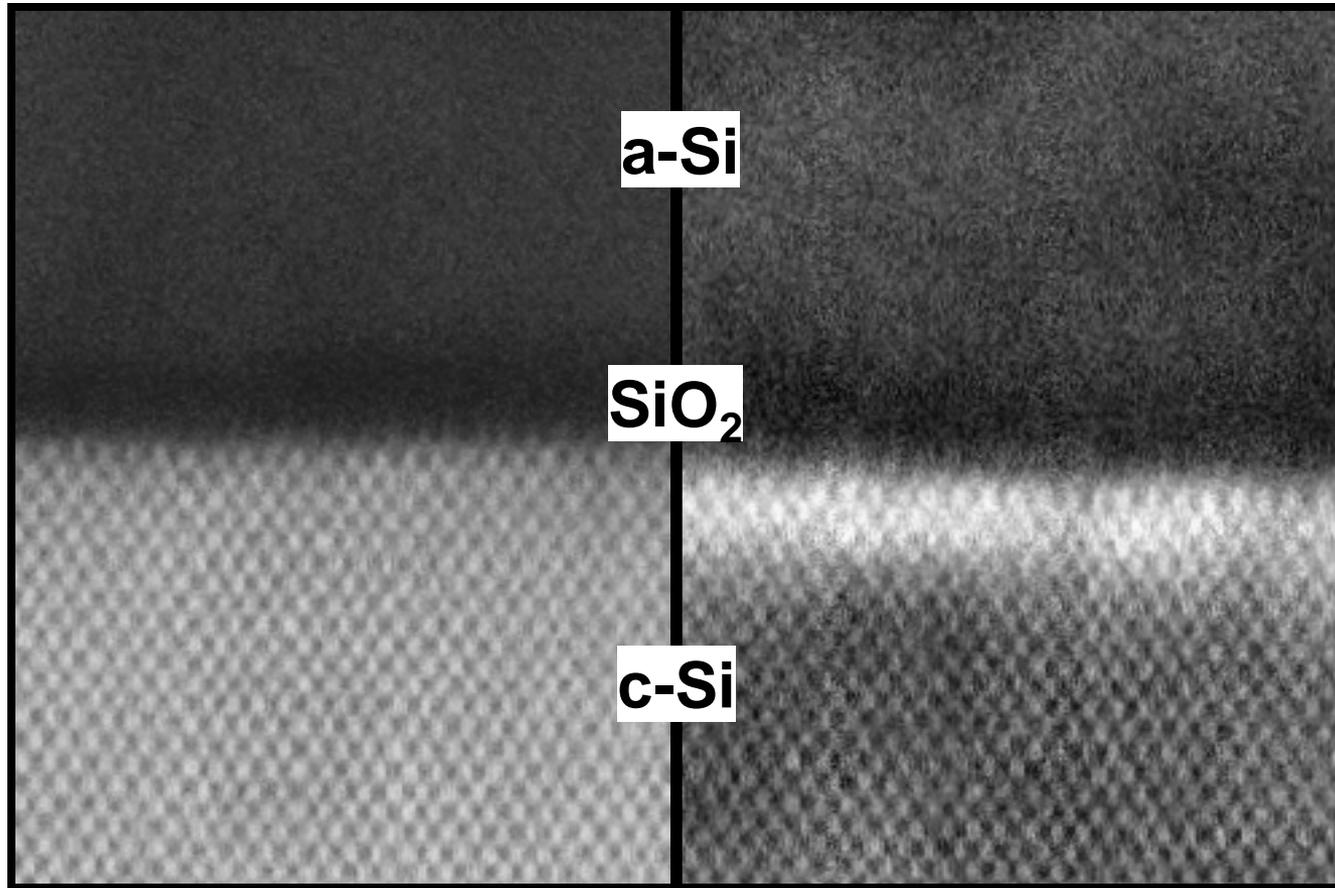
(JEOL 2010F, 200 kV, C<sub>s</sub>=1mm)

---

ADF Inner angle:

50 mrad

25 mrad



Strain Fields cause dechanneling (and scattering to small angles)

Z. Yu, D. A. Muller, and J. Silcox, *J. Appl. Phys.* **95**, 3362 (2004).

# When Does the Simple Picture Work?

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- Simple incoherent,  $Z^{1.7}$  theory works well:
  - for an amorphous specimen
  - for randomly-oriented polycrystalline specimens
  - (mostly) for single crystals tilted between major zone axes
- It doesn't work for single crystals oriented on a zone axes – i.e. when you get atomic-place resolution.